

## Crystal Chemistry of the VA Element Trihalides: Lone Pair, Stereochemistry, and Structural Relationships\*

JEAN GALY AND RENÉE ENJALBERT

*Laboratoire de Chimie de Coordination du CNRS, 205 route de Narbonne, 31400 Toulouse, France*

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The crystal structures of  $\text{NCl}_3$ ,  $\text{PCl}_3$ ,  $\text{PBr}_3$ ,  $\text{AsF}_3$ , and  $\text{AsCl}_3$  liquids at room temperature have been recently determined. The data for the last two are reported in the present paper. The structures of the following trihalides are discussed and classified in structural types:  $\text{AsF}_3$ ,  $\text{SbF}_3$ ,  $\alpha\text{BiF}_3$ ,  $\text{YF}_3$  ( $\text{NCl}_3$ ,  $\text{PCl}_3$ ,  $\text{PBr}_3$ ,  $\beta\text{SbCl}_3$ ,  $\beta\text{SbBr}_3$ ,  $\gamma\text{BiF}_3$ , and  $\beta\text{BiCl}_3$ ),  $\text{AsBr}_3$  ( $\text{AsCl}_3$ ,  $\text{AsBr}_3$ , " $\alpha\text{SbCl}_3$ ," and  $\alpha\text{SbBr}_3$ ),  $\text{PI}_3$  and  $\alpha\text{AsI}_3$  ( $\alpha\text{AsI}_3$ ,  $\text{SbI}_3$ , and  $\text{BiI}_3$ ). Structural relationships have been established between  $\text{AsF}_3$  and  $\text{XeO}_3$ ,  $\text{PX}_3E$  and  $\text{POX}_3$  ( $X = \text{Cl}, \text{Br}$ ), and  $\text{AsBr}_3E$  has been compared with  $\text{PuBr}_3$  and  $\text{VOCl}_3$ . A structural mechanism is proposed for the polymorphic transformation  $\alpha\text{SbBr}_3 \rightleftharpoons \beta\text{SbBr}_3$ . The  $\text{CrCl}_3$  structure type is proposed for the high-temperature form of  $\text{AsI}_3$  and a mechanism is given for the phase transition  $\alpha\text{AsI}_3 \rightleftharpoons \beta\text{AsI}_3$ . A proposal is made for a possible high-pressure transformation of the  $\alpha\text{AsI}_3$  structure into a  $\text{VF}_3$  structure type. The stereochemistry is discussed in terms of coordination polyhedra around  $M^*$ ; all these trihalides are characterized by a basic molecular unit  $M^*X_3$ . The steric effects of the lone pair  $E$  carried by  $M^*$  is especially emphasized and discussed; two characteristics of  $E$  are presented: the centroid of the electronic doublet located by  $E_c$  and the sphere of influence located by its center  $E_s$ . The correlative variations of  $M^*-E_c$  and  $M^*-E_s$  are studied according to the nature of  $M^*$  ( $\text{N(III)}$ ,  $\text{P(III)}$  to  $\text{Cl(V)}$ ,  $\text{Ga(I)}$  to  $\text{Br(V)}$ ,  $\text{In(I)}$  to  $\text{Xe(VI)}$ , and  $\text{Tl(I)}$  to  $\text{Po(IV)}$ ).

### Introduction

The chemistry and stereochemistry of oxides containing transition elements and elements  $M^*$  carrying a lone pair  $E$  were an extremely attractive field for one of us (JG) because of the wide variety offered of new compounds apt to be formed and because of the original structures such compounds exhibit. Such an area was investigated through the chemistry based on tellurium(IV) possessing the electronic structure  $\{\text{Kr}\}4d^{10}5s^2$ . The stereochemical influence of the lone pair on tellurium(IV) was

readily enhanced and the various crystal structures compared with those "lone pair" elements like  $\text{Sb(III)}$ ,  $\text{Pb(II)}$ , etc. (1). The various coordination polyhedra of tellurium were investigated by Zemann (2) who gave a most valuable contribution to the understanding of tellurium(IV) structural chemistry. A further exhaustive contribution to this field was given by Meunier (3).

The lone-pair steric effects have been analyzed in terms of interelectron repulsion by Sidgwick and Powell (4). Their theory was modified a few years later by Gillespie and Nyholm (5) who assumed lone pairs to be larger than bonding pairs, so that the repulsion between electron pairs decreases as: lone pair-lone pair > lone pair-bonding

\* Dedicated to Professor A. F. Wells on his 70th birthday.

pair > bonding pair–bonding pair (VSEPR theory). The electron pairs are at the corner of a polyhedron with the nucleus at its center, the considerable distortion occurring due to the larger size of the lone pair, sitting in the site normally occupied by a ligand, according to the previous authors. An important and fundamental investigation by *ab initio* calculations was recently performed by Schmiedekamp *et al.* (6) on the VSEPR models.

In the area of structural chemistry, a fundamental observation was pointed out by Andersson and Åström (7) working in the field of oxides, fluorides, and oxide fluorides of “ $ns^2$  elements”: the lone pair in these compounds takes the same volume as an oxide  $O^{2-}$  or a fluoride  $F^-$  ion. This is easily checked by calculating the volume per anion in “lone pair” compounds. This volume is of the same importance as the one found in transition metal oxides, i.e., 16–18 Å<sup>3</sup>, if the lone pairs are taken into account (only true for elements with a single lone pair).

We named the space required by the non-bonded electron pair “the sphere of influence of the lone pair  $E$ ,” which may include the associated cation. Hence the center of this sphere, referred to as  $E_s$ , was placed at the vertex of a regular polyhedron, such as a tetrahedron, a trigonal bipyramid, or an octahedron. Then, following these principles, Galy *et al.* (8) derived the distance nucleus– $E_s$  for the  $M^*$  elements, Ge(II) to Br(V), Sn(II) to Xe(VI), and Tl(I) to Bi(III), after a thorough study of a number of oxide and fluoride structures. The distance nucleus–centroid of the lone pair, i.e.,  $E_c$ , is consequently somewhat shorter than the distance nucleus– $E_s$ .

In order to extend this analysis of the stereochemical effects of the lone pair  $E$  to other ligands fixed on  $M^*$  elements, an investigation of the crystal structures of the trihalides of VA elements was attempted. Some of them were not known or poorly

defined. A few structures were then determined and refined and are reported in the present paper, together with the results found in the literature.

All these trihalides  $M^*X_3$  are characterized in the gas, liquid, and solid states by a simple pyramidal molecular structure. However, in the solid state, the crystal network exhibits various molecular packings. In Sections I, II, and III of the present study, we will deal with the classification by structural type, with the aim of reporting the general organization of the packing and the structural relationships, and describing the various coordination polyhedra in which the element  $M^*$  and its lone pair adopt. The last section will report the characteristics of  $M^*X_3$  molecules and the stereochemical effects of the lone pair  $E$  on the structures.

### I. $M^*F_3$ Trifluorides

The first two trifluorides of this series,  $NF_3$  and  $PF_3$ , are gaseous at room temperature. The structures of these molecules have only been studied by electron diffraction.  $AsF_3$ ,  $SbF_3$ , and  $BiF_3$  have different structural types, the later showing various polymorphic forms.

#### $AsF_3$ —A Structural Type

The detailed structure of crystalline  $AsF_3$ , liquid at room temperature, was reported by Enjalbert and Galy (9) (Table I). The detailed experimental data are developed in Enjalbert's thesis (10), some results of which are reported in Appendix I.

$AsF_3$  crystallizes in the orthorhombic system and the atomic arrangement is shown in Fig. 1 (noncentrosymmetric space group  $Pn2_1a$ ). The  $AsF_3$  unit with its arsenic and three fluorine atoms has approximately the expected  $C_{3v}$  symmetry. However, there are six other fluorine neighbors completing a distorted tricapped trigonal prism around  $AsE$ .

TABLE I  
CRYSTALLOGRAPHIC DATA FOR MOLECULES  $M^*X_3$

Com- pound	Molec- ular weight	Melt- ing point (K)	Liquid specific mass (g cm <sup>-3</sup> )	Crystal specific mass (g cm <sup>-3</sup> )RX	Temper- ature of X-ray study (K)	Crystal system	Space group	(Unit cell parameters)			Vol- ume (Å <sup>3</sup> )	Number of molecules/cell	Reliability factor of the structure determination	Refer- ence
								a (Å)	b (Å)	c (Å)				
NF <sub>3</sub>	71.0	66	—	—	—	—	—	—	—	—	—	—	—	—
PF <sub>3</sub>	88.0	121	—	—	—	—	—	—	—	—	—	—	—	—
AsF <sub>3</sub>	131.9	267	2.666	3.28	193	Orth.	<i>Pn2<sub>1</sub>a</i>	7.018(5)	7.315(6)	5.205(6)	267	4	0.055	(9)
SbF <sub>3</sub>	178.8	565	—	4.39	293	Orth.	<i>Ama</i> 2	7.26(1)	7.46(1)	4.95(1)	268	4	0.072	(13)
αBiF <sub>3</sub>	266.0	1000	—	8.76	293	Cubic	<i>P4<sub>3</sub>m</i>	5.865(6)	—	—	202	4	—	(15)
γBiF <sub>3</sub>	266.0	1000	—	7.88	293	Orth.	<i>Pnma</i>	6.56	7.03	4.86	224	4	—	(17)
δBiF <sub>3</sub>	—	—	—	—	—	Tetra.	—	7.076	—	7.350	—	—	—	(16)
NCl <sub>3</sub>	120.4	<233	1.653	2.08	148	Orth.	<i>Pnma</i>	7.48(2)	9.35(1)	16.48(2)	1153	12	0.091	(33)
PCl <sub>3</sub>	137.3	161	1.574	1.99	123	Orth.	<i>Pnma</i>	8.0425(8)	9.3691(7)	6.0794(4)	458	4	0.029	(21)
AsCl <sub>3</sub>	181.3	264	2.163	2.62	253	Orth.	<i>P2<sub>1</sub>2<sub>1</sub>2<sub>1</sub></i>	9.466(3)	11.335(3)	4.289(2)	460	4	0.036	(19)
βSbCl <sub>3</sub>	228.1	346	—	3.14	293	Orth.	<i>Pnma</i>	8.111(2)	9.419(1)	6.313(1)	482	4	0.045	(31)
αBiCl <sub>3</sub>	315.3	503	—	3.89	293	Cubic	<i>P2<sub>1</sub>3</i>	8.14	—	—	539	4	—	(41)
βBiCl <sub>3</sub>	315.3	503	—	4.75	293	Orth.	<i>Pnma</i>	7.641(2)	9.172(7)	6.291(2)	441	4	0.044	(32)
NBr <sub>3</sub>	—	—	—	—	—	—	—	—	—	—	—	—	—	—
PBr <sub>3</sub>	270.7	233	2.852	3.47	193	Orth.	<i>Pnma</i>	8.014(4)	10.026(8)	6.444(4)	518	4	0.034	(20)
AsBr <sub>3</sub>	314.7	306	—	3.86	293	Orth.	<i>P2<sub>1</sub>2<sub>1</sub>2<sub>1</sub></i>	10.240(5)	12.200(5)	4.33(1)	541	4	0.19	(37)
αSbBr <sub>3</sub>	361.5	370	—	4.37	293	Orth.	<i>P2<sub>1</sub>2<sub>1</sub>2<sub>1</sub></i>	10.12(1)	2.30(1)	4.42(1)	550	4	0.18	(38)
βSbBr <sub>3</sub>	361.5	370	—	4.38	293	Orth.	<i>Pnma</i>	8.25(1)	9.96(1)	6.68(1)	549	4	0.104	(30)
αBiBr <sub>3</sub>	448.7	491	—	3.79	293	Cubic	<i>P2<sub>1</sub>3</i>	9.23	—	—	786	4	—	(41)
N <sub>2</sub>	394.7	—	—	—	—	—	—	—	—	—	—	—	—	—
PI <sub>3</sub>	411.7	334	—	4.18	293	Hexag.	<i>P6<sub>3</sub></i>	7.133(2)	—	7.414(2)	327	2	0.065	(44)
αAsI <sub>3</sub>	455.6	419	—	4.74	293	Trig.	<i>R3</i>	7.193(2)	—	21.372(7)	958	6	0.033	(46)
βAsI <sub>3</sub>	455.6	419	—	4.71	393	Trig.	<i>P3<sub>1</sub>12</i>	7.20(5)	—	21.47(8)	964	6	—	—
SbI <sub>3</sub>	502.5	443	—	4.94	293	Trig.	<i>R3</i>	7.48(2)	—	20.90(5)	1013	6	0.131	(49)
BI <sub>3</sub>	589.5	681	—	5.81	293	Trig.	<i>R3</i>	7.514(2)	—	20.702(7)	1012	6	0.15	—

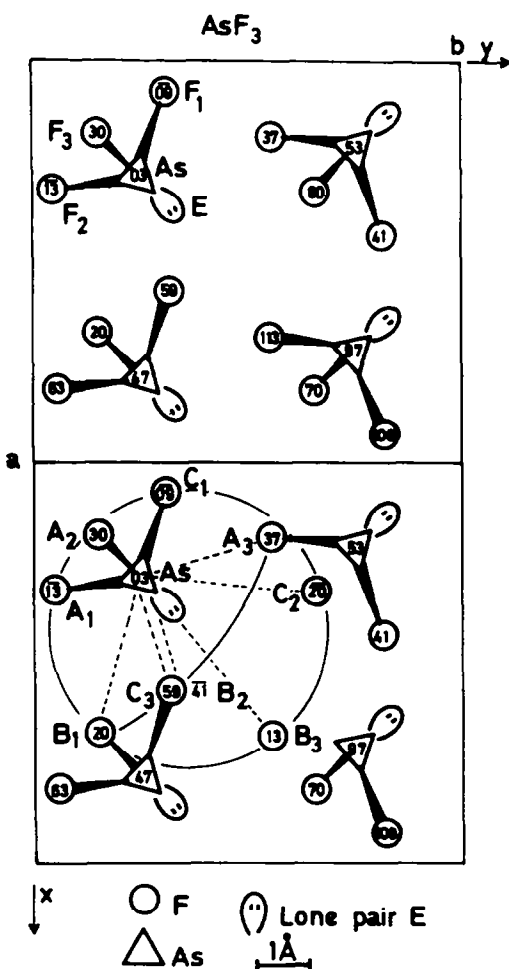


FIG. 1. Projection of the  $\text{AsF}_3$  crystal structure onto the (001) plane. Coordination polyhedron around  $\text{AsE}$ .

If we draw a schematic polyhedron packing of  $\text{AsF}_3$  with the trigonal prisms surrounding the lone pair  $E$ , Fig. 2a is obtained showing layers of alternated prisms along the  $a$  axis; an important capping summit is indicated to remind us that an arsenic atom of a layer of trigonal prisms is connected to a fluorine of the neighboring unit.

*Structural relationship between  $\text{AsF}_3$  and  $\text{XeO}_3$ .*  $\text{XeO}_3$  is orthorhombic ( $a = 6.16$ ,  $b = 8.11$ ,  $c = 5.23$  Å, space group  $P2_12_12_1$ ) (11). Andersson (12) discussed the stereochemistry of valence bonds and the struc-

ture of the xenon trioxide together with those of the fluorides  $\text{XeF}_4$  and  $\text{XeF}_2$ .

$\text{Xe(VI)}$ , like  $\text{As(III)}$ , possesses a lone pair. In Fig. 2b (drawn according to Andersson (12)), it is readily seen that, despite some distortion,  $\text{XeO}_3$  shows a structural packing remarkably similar to that of  $\text{AsF}_3$ .

#### $\text{SbF}_3$ —A Structural Type

The crystal structure of  $\text{SbF}_3$  was reinvestigated by Edwards (13) (Table I). The  $\text{SbF}_3$  units are linked through three fluorine bridges to form a three-dimensional net-

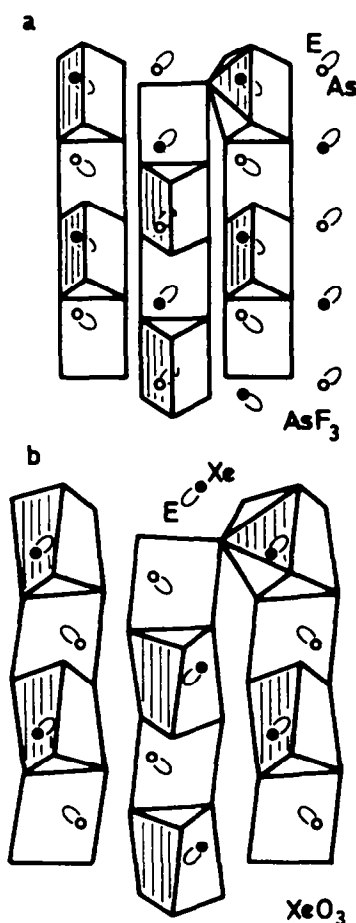


FIG. 2.  $\text{AsF}_3$  and  $\text{XeO}_3$  crystal structure. Small circles are  $\text{As(III)}$  or  $\text{Xe(VI)}$  situated in the capped square pyramid; lone pairs are located in the trigonal prisms.

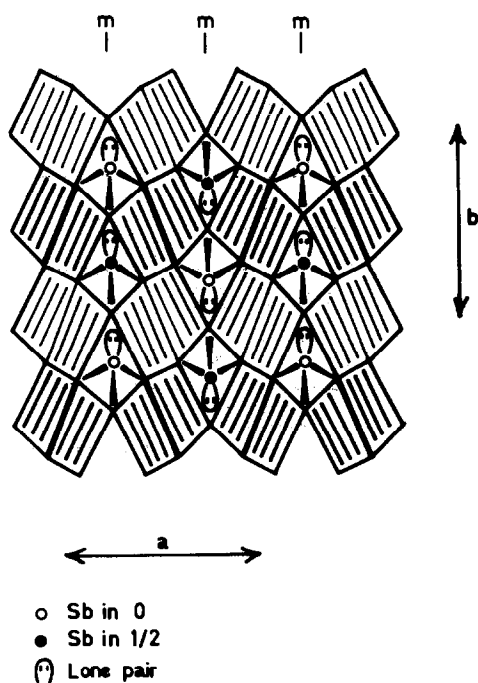


FIG. 3. Packing of fluorine rhombic prisms between the mirror planes in the  $\text{SbF}_3$  crystal structure.

work with a distorted trigonal antiprism of fluorine around  $\text{SbE}$  which reside together with fluorine atoms in mirror planes. The general structure can be described as being composed of layers of fluorine rhombic prisms sharing edges and vertices. The mirrors play the role of "twinning" planes accommodating the  $\text{Sb}$  atoms. The  $\text{SbF}_3$  structure, slightly idealized in this way, is represented in Fig. 3.

### $\text{BiF}_3$

Two polymorphs of this ionic fluoride,  $\alpha$  and  $\gamma$ , have been structurally characterized.

$\alpha\text{BiF}_3$ .  $\alpha\text{BiF}_3$  crystallizes in the cubic system (Table I) and gives a polymorphic  $\beta$  form above a transition temperature of 473 K (14). According to Hund and Fricke (15), in the  $\alpha\text{BiF}_3$  structure  $\text{Bi}$  atoms should be surrounded by eight fluorine atoms forming a cube. In fact, a displacement of bismuth

toward one fluoride along the threefold axis of the cube decreases the coordination number to 7 and the fluorine polyhedron around  $\text{Bi}$  becomes a monocapped trigonal antiprism.

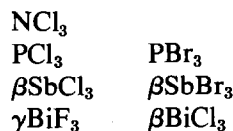
$\gamma\text{BiF}_3$ . The orthorhombic  $\gamma$  form (Table I) pertains to an isostructural family including several  $M^*\text{Cl}_3$  and  $M^*\text{Br}_3$  compounds and classified as a  $\text{YF}_3$  structure type described below.

$\delta\text{BiF}_3$ . This phase appears at 5–7 kbars and 20°C when the  $\gamma$  form undergoes shock compression (16); it is tetragonal with  $a = 7.076$ ,  $c = 7.350$  Å (Table I).

## II. $M^*\text{X}_3$ Trichlorides and Tribromides

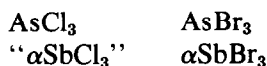
These trihalides can be assigned two main structure types, respectively:

(1) the  $\text{YF}_3$  crystal structure:



and

(2) the  $\text{AsBr}_3$  crystal structure:



The bismuth compounds exhibit a cubic form, the crystal structure of which has not been determined.

### $\text{YF}_3$ Structure Type

The  $\text{YF}_3$  structure has been determined by Zalkin and Templeton (17) (orthorhombic, space group  $Pnma$ ,  $a = 6.353$ ,  $b = 6.850$ ,  $c = 4.393$  Å) and can be classified as an anti- $\text{Fe}_3\text{C}$  type. In the cementite structure, described as an example of chemical twinning in HCP, one  $\text{C}$  atom is situated in a two-capped trigonal prism of  $\text{Fe}$  atoms (18).

In  $\text{YF}_3$  and in their related  $M^*\text{X}_3$  the cations, as well as the lone pair, are situated inside the prisms. The idealized picture of this typical structure is given in Fig. 4. In



$\beta\text{SbBr}_3$  (30). A refinement of  $\beta\text{SbCl}_3$  has been recently published by Lipka (31). These two halides belong to the  $\text{YF}_3$  structure type.

$\gamma\text{BiF}_3$  and  $\beta\text{BiCl}_3$ .  $\gamma\text{BiF}_3$  is isostructural with  $\text{YF}_3$ , as indicated by Zalkin and Templeton (17).

From a detailed X-ray single crystal study, Nyburg *et al.* (32) claimed  $\beta\text{BiCl}_3$  belonged to the  $Pn2_1a$  space group instead of the most symmetrical  $Pnma$ . They obtained a very small distortion and, in spite of Hamilton's test, the result supporting  $Pn2_1a$  is not very convincing (especially in a crystal structure containing heavy elements such as bismuth) (Table I). But the NQR study (23) definitively resolved this problem by showing only two signals, which then implies the  $Pnma$  space group.

$\text{NCl}_3$ —An unexpected superstructure. This compound, investigated by Hartl *et al.* (33), crystallizes in the orthorhombic system, space group  $Pnma$ , with the  $c$  parameter three times as large as that in the  $\text{YF}_3$  structure type (Table I). Consequently, there are three crystallographically independent nitrogen atoms in the cell ( $\text{N}(1)$ ,  $\text{N}(2)$ ,  $\text{N}(3)$ ) and the  $\text{NCl}_3$  molecules still contain the mirror plane as a symmetry element. The packing of these molecules is such that  $\text{N}(1)$  is located in a large trigonal antiprism and  $\text{N}(2)$  and  $\text{N}(3)$  in bicapped trigonal prisms.

A projection on the mirror plane has been prepared to compare this structure with the previous ones.  $\text{PBr}_3$  is represented in Fig. 6 and  $\text{NCl}_3$  in Fig. 7. These figures clearly show the networks formed by the trigonal prisms in  $\text{PBr}_3$  and the trigonal prisms and antiprisms in  $\text{NCl}_3$ . It is seen that the tripling of the  $c$  parameter in  $\text{NCl}_3$  is due to the tilting of the  $\text{N}(1)\text{Cl}_3$  molecule in the same direction as  $\text{N}(2)\text{Cl}_3$  along the line of chlorine atoms following the direction  $\{100\}$ . The  $\text{N}(3)\text{Cl}_3$  molecules are packed in a way more similar to  $\text{PBr}_3$ , for example. These figures, especially Fig. 6, directly

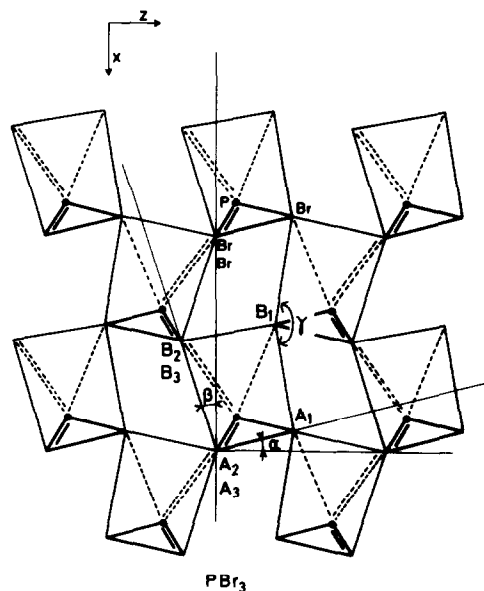


FIG. 6. A layer of  $M^*X_3$  molecules; projection onto the mirror plane ( $\text{PBr}_3$  structure) with the  $M^*$  atoms in trigonal prisms;  $\alpha$ ,  $\beta$ , and  $\gamma$  angles indicate the general distortion of the packing.

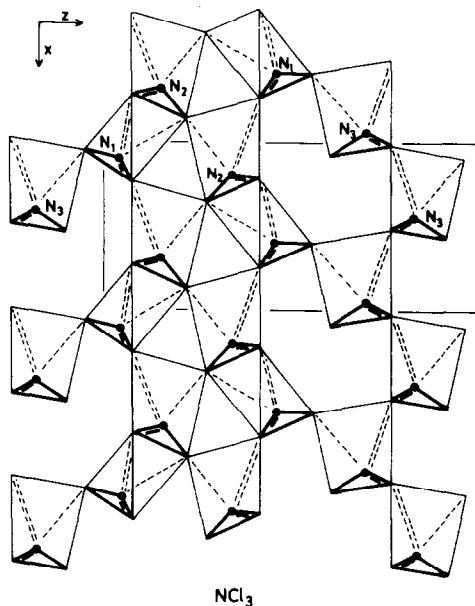


FIG. 7. A layer of  $\text{NCl}_3$  molecules; it shows the combination of chlorine trigonal prisms (around  $\text{N}(2)$  and  $\text{N}(3)$  atoms) and octahedra (around  $\text{N}(1)$ ) leading to the tripling of the  $c$  parameter.

concern the whole series of  $YF_3$  structure type compounds and they show how distorted the packing of the trigonal prisms can be.

To evaluate this distortion, we defined the angles  $\alpha$  = dihedral angle between the triangle built of the halogens of the  $M^*X_3$  molecule and the plane (100),  $\beta$  = dihedral angle between the rectangular face of the trigonal prism composed of the halogens related by the mirror plane and the plane (001), and  $\gamma$  = the angle between successive halogens belonging to the mirror plane in the {100} direction. These angles are presented in Table II. The corresponding distortion for  $POCl_3$  and  $POBr_3$  are included for comparison.

#### *AsBr<sub>3</sub> Structure Type*

This structure was determined in 1935 on a single crystal of arsenic tribromide by Braekken (34).

*AsCl<sub>3</sub> and AsBr<sub>3</sub>.* The crystal structure of  $AsCl_3$  has been investigated at low temperature by Enjalbert and Galy (10,19) (cf. Appendix II). A drawing of the molecular disposition is given in Fig. 8. The molecules are piled up in the {001} direction as infinite trigonal chains and the arsenic atoms are located in the trigonal prisms. All these

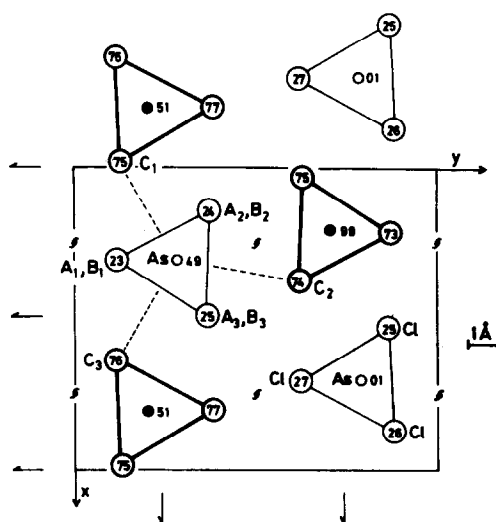


FIG. 8. Detailed projection of the  $AsCl_3$  structure onto the plane (001).

chains are parallel, which brings three extra chlorine atoms close to the arsenic and its lone pair. A coordination polyhedron is thus defined as a tricapped trigonal prism.

The  $AsBr_3$  structure has been redetermined by Singh and Swaminathan (35) and Trotter (36) and refined ( $R = 0.143$ ) by Singh and Swaminathan (37).

" $\alpha SbCl_3$ " and  $\alpha SbBr_3$ . The compound " $\alpha SbCl_3$ " is reported by Lindqvist and Niggli (29) but no X-ray work has been cited.

The structure  $\alpha SbBr_3$  has been determined by Cushen and Hulme (38).

*Structural relationships with  $PuBr_3$  and  $VOCl_3$ .* The typical infinite trigonal chains of the  $M^*X_3$  compounds with the  $AsBr_3$  structure type are held together by weak interactions and an idealized drawing is given in Fig. 9.

$PuBr_3$  is built up of similar chains (orthorhombic, space group  $Ccmm$ ,  $a = 12.64$ ,  $b = 4.10$ ,  $c = 9.14$  Å (39)) but, despite similar tricapped trigonal prism coordination polyhedra around the plutonium atoms, the general packing appears different (Fig. 10). Yet, the structure of  $VOCl_3$

TABLE II  
DISTORTION OF THE  $M^*X_3$  PACKING IN  $YF_3$   
RELATED STRUCTURES<sup>a</sup>

Compound	$ \alpha $	$ \beta $	$ \gamma $
$PCl_3$	17	0	150
$PBr_3$	13	19	160
$\beta SbCl_3$	16	20	152
$\beta SbBr_3$	16	19	151
$\beta BiCl_3$	17	20	144
$N(1)Cl_3$	27	36	153
$NCl_3$ $N(2)Cl_3$	27	0	154
$N(3)Cl_3$	10	0	153
$POCl_3$	26	13	136
$POBr_3$	26	12	140

<sup>a</sup>  $\alpha$ ,  $\beta$ , and  $\gamma$  in degrees.



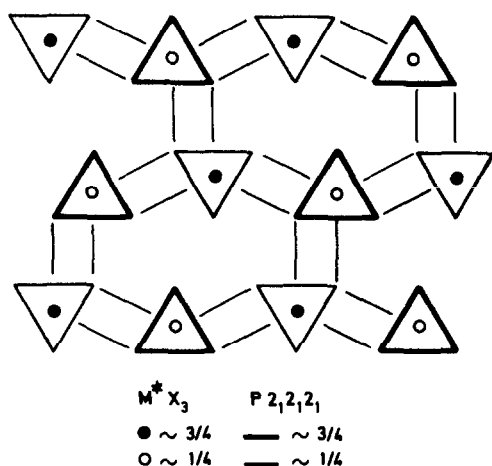


FIG. 9. Ideal packing of the infinite linear chains of  $M^*$  filled halogen trigonal prisms with  $AsBr_3$  structure type (triangles in projection onto the plane (001)). Thin lines indicate the weak secondary interchain bonding, yielding a tricapped trigonal prism coordination around  $M^*$ .

recently investigated in our laboratory by Enjalbert *et al.* (40) (orthorhombic, space group  $Pnma$ ,  $a = 4.963(1)$ ,  $b = 9.140(4)$ ,  $c = 11.221(5)$  Å at 133 K) seems more closely related to  $AsBr_3$ . The packing is derived from  $AsBr_3$  by a slip affecting every two layers of chains parallel to the plane (100)

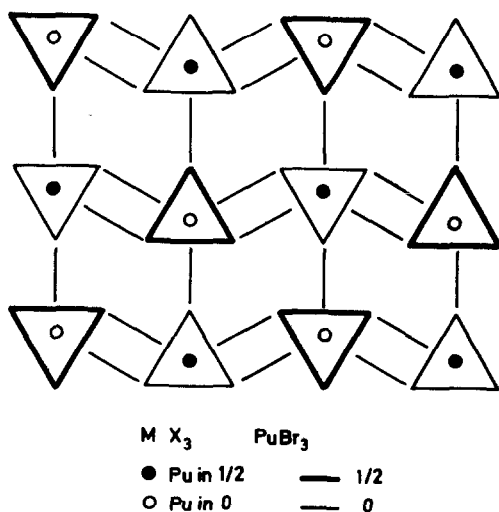


FIG. 10. Representation of the  $PuBr_3$  structure in a way similar to Fig. 9.

and by the adjusting of the trigonal chains on the mirror planes (Fig. 11). The VO group is inserted in a bicapped trigonal prism of chlorine atoms.

A possible mechanism for the  $\alpha SbBr_3 \rightleftharpoons \beta SbBr_3$  transformation. The structure of  $\alpha SbBr_3$  is represented in Fig. 12a. When each layer B is moved in the direction of the arrow by a vector  $Br-Br$ , the arrangement drawn in Fig. 12b is obtained. As indicated in the right portion of the drawing, the infinite trigonal chains containing the antimony atoms and their lone pairs are now separated by empty infinite trigonal chains (indicated with dashed lines in the projection). Half of the antimony atoms may move easily through the rectangular face of trigonal prisms into the opposite empty ones (indicated by arrows). The detail of this operation is further shown in Fig. 13. The parameter along the chains is then doubled and the structure of  $\beta SbBr_3$  is obtained (Fig. 14).

The  $\alpha$  form of  $BiCl_3$  and  $BiBr_3$ . According to Wolten and Mayer (41),  $\alpha BiCl_3$  and

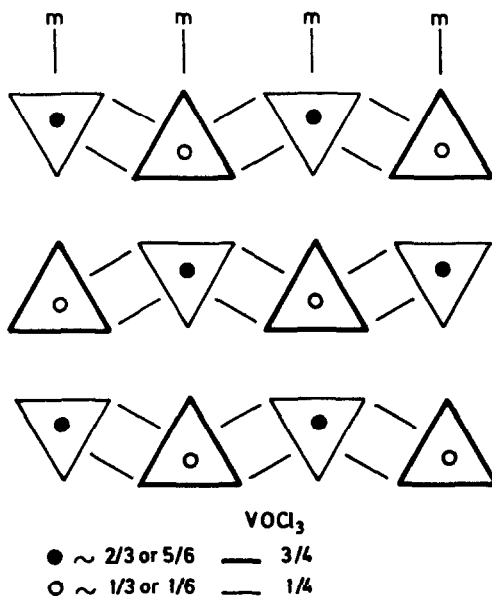


FIG. 11. Representation of the  $VOCl_3$  structure as in  $AsBr_3$  and  $PuBr_3$  structure types.



These structures are better understood using the schematic representation developed by Andersson and Galy (42) describing Wadsley's defects and crystallographic shears in hexagonally close packed structures. Figure 15 shows the different ways of joining the octahedra in a HCP arrangement.

### $PI_3$ —A Structural Type

The crystal structure of phosphorus triiodide was originally investigated by Braekken (43) who determined the iodine atom positions. However, the phosphorus positions were not resolved. Wyckoff (14) indicated that the phosphorus atom, similarly to the boron of the isostructural compounds  $BCl_3$ ,  $BBr_3$ , and  $BI_3$ , is located in the middle of the halide triangle. Such an assertion is unreasonable in the case of phosphorus(III), since the size of the site is too small to accommodate the P atom with its lone pair. This assignment is also unrealistic for the molecular structure.

The crystal structure of  $PI_3$  was reinvestigated by Lance *et al.* (44). The phosphorus atoms were shown to be situated above the planes of a nearly ideal HCP of iodine atoms, and the pyramidal geometry of the

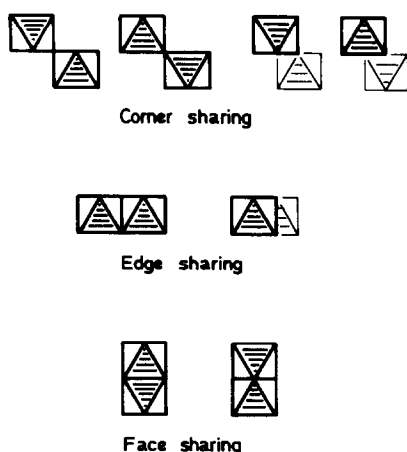


FIG. 15. A few possible ways of joining octahedra when the projection axis is  $a + b$  in the hexagonally close packed arrangement of iodine atoms.

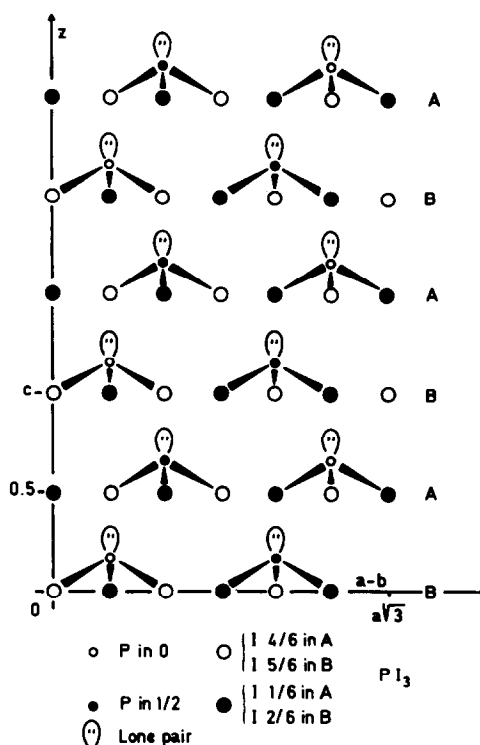


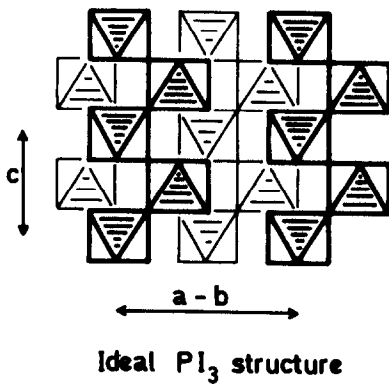
FIG. 16. The projection of the  $PI_3$  structure onto the plane (110).

molecule is maintained inside the solid (Fig. 16, Table I). The stereochemical influence of the lone pair is clearly indicated: the structure is isostructural with triiodomethane (45). The phosphorus atom and its lone pair are located in the octahedral iodine site. These octahedra appear as some kind of zig-zag chains along the  $c$  axis connected to each other by sharing adjacent corners (Fig. 17).

Among the  $M^*X_3$  structure types, both  $PI_3$  and  $VF_3$  constitute examples where only one-third of the octahedral holes of HCP are occupied. The ideal  $VF_3$  structure type is given in Fig. 18 to facilitate the comparison (14, 42).

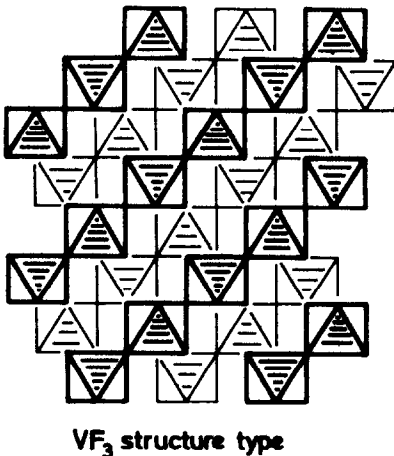
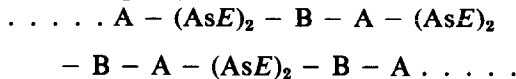
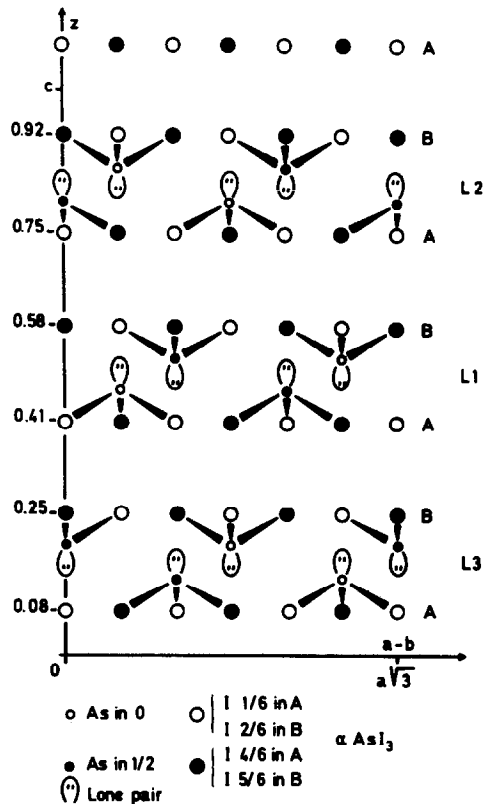
### $\alpha AsI_3$ —A Structural Type

$AsI_3$ —the  $\alpha$  form. The refinement ( $R = 0.033$ ) of the crystal structure of  $\alpha AsI_3$  has been recently reported (46). In 1931, Hey-

FIG. 17. The  $PI_3$  structure type.

worth (47) published the original structure with pertinent comments concerning the arsenic stereochemistry. Later on, Trotter's work (48) confirmed both the structure and the remarks in spite of the poor reliability factors obtained ( $R = 0.11$  and  $R = 0.16$  for  $hk0$  and  $h0l$  film data).

The structure can be described as constituted of discrete  $AsI_3$  molecules; the As atoms and their lone pair  $E$  are located in two out of three octahedra every two layers (A, B) of the hexagonal close packing of iodine atoms (Fig. 19):

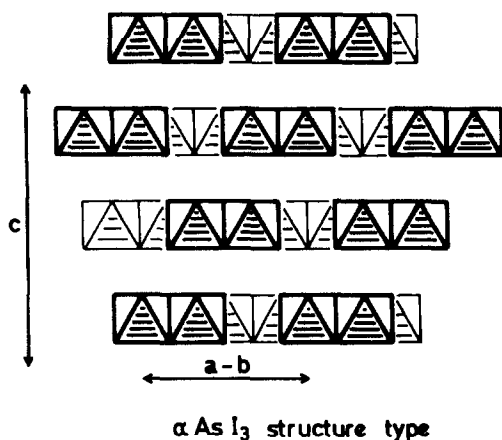
FIG. 18. The  $VF_3$  structure type.FIG. 19. The projection of  $\alpha AsI_3$  structure onto the plane (110).

The ideal drawing of the  $\alpha AsI_3$  structure (Fig. 20) shows the arrangement of filled iodine octahedra sharing edges and corners.

$SbI_3$ . The crystal structure determination of  $SbI_3$  was performed by Trotter and Zobel (49) using X-ray photographic data.  $SbI_3$  is isostructural with  $AsI_3$  (Table I).

$BiI_3$ .  $BiI_3$  is commonly referred to in the literature as a structural type, which obviously overlooks the original work on the  $\alpha$  form of  $AsI_3$  by Heyworth (47).

Nevertheless, from the powder pattern analysis, Trotter and Zobel (49) claimed the bismuth(III) to be in the center of the iodine octahedron which implied that the molecular form  $BiI_3$ , as in the preceding iodides, was no longer present in the solid.

Fig. 20. The  $\alpha \text{AsI}_3$  structure type.

Our own interest in the "lone pair element" stereochemistry prompted us to re-examine this question. We started growing single crystals of  $\text{BiI}_3$  and studying them by X-ray techniques. Partial results have now been obtained concerning its crystal structure (crystal data in Table I) which may be briefly summarized: (1) the tripling of the  $c$  axis is confirmed although some authors report  $a = 7.50$ ,  $c = 6.9 \text{ \AA}$  (14), (2) the iodine packing is HCP, and (3) the calculations indicate that the bismuth atoms seem to occupy most of the octahedral sites in a non-stoichiometric way, in spite of a preference for the  $\alpha \text{AsI}_3$  distribution type of atoms.

Further work is needed, to control the crystal growing and annealing. The possibility of an ordered phase with the  $c$  parameter ca.  $7 \text{ \AA}$  and of another phase with a triple  $c$  value, as well as of a multiple twinning between the two phases, cannot be excluded.

### Polymorphism of $\text{AsI}_3$

The high temperature form  $\text{AsI}_3$ .  $\alpha \text{AsI}_3$ , in the form of a deep orange powder, when heated at ca.  $110^\circ\text{C}$  in a Pyrex tube yields a pale yellow crystalline phase in the cooler portion of the tube. After a few days at room temperature, this phase turns orange. It obviously transformed back to orange

$\alpha \text{AsI}_3$ . This "high"-temperature form of  $\text{AsI}_3$ , already reported by Horiba and Inouye (50), is called  $\beta \text{AsI}_3$ .  $\text{AsI}_3$  therefore exhibits a reversible polymorphic transformation:  $\alpha \text{AsI}_3 \xrightleftharpoons{110^\circ\text{C}} \beta \text{AsI}_3$ .

*X-ray powder analysis.* The experiment above was repeated in a Lindemann capillary and a powder pattern of the yellow phase was recorded using the Debye-Scherrer technique ( $\text{CuK}\alpha$  radiation). The interplanar distances of the observed diffraction lines, together with the visually estimated intensities, are summarized in Table III.

*$\beta \text{AsI}_3$  structure—a possible  $\text{CrCl}_3$  type?* The  $\beta \text{AsI}_3$  powder pattern is different from that of  $\alpha \text{AsI}_3$ ; nevertheless it can be indexed using an analogous hexagonal cell, with the following parameters:  $a = 7.20$  and  $c = 21.47 \text{ \AA}$ , but the extinction rules are quite different, the major ones being  $00l = 3n + 1$ .

From a thorough NQR study on the  $\beta$  phase by Kojima *et al.* (51), this crystalline phase is claimed to contain three crystallographically independent iodine atoms. Up to 383 K, only one signal is obtained ( $\nu_1^{27}\text{I} = 207.023 \text{ MHz}$ ), showing the three iodine atoms of the molecule to be crystallographically equivalent in the  $\alpha \text{AsI}_3$  structure type. Above this temperature, three distinct signals appear ( $\nu_1^{27}\text{I} = 206.20$ ,  $208.45$ , and

TABLE III  
 $\text{AsI}_3$  POWDER PATTERN

$l_{\text{obs.}}$	$d_{\text{obs.}} (\text{\AA})$	$d_{\text{calc.}} (\text{\AA})$	$hkl$
w	3.57	3.578	0 0 6
w	3.40	3.413	1 1 2
vS	3.225	3.216	1 1 3
w	3.016	2.994	0 2 2
		2.990	1 1 4
S	2.554	2.538	1 1 6
S	2.078	2.078	0 3 0
w	1.970	1.968	1 2 6
mS	1.789	1.789	0 0 12

211.74 MHz), supporting the conclusions made by these authors (51).

The answer to the  $\beta\text{AsI}_3$  structural problem could then be assumed by the  $\text{CrCl}_3$  structure type determined by Natta (52) and refined by Wooster (53), which requires cell parameters similar to  $\alpha\text{AsI}_3$  but with space group  $P3_212$  (or its enantiomorphic form  $P3_112$ ) which imposes the extinction  $00l \mid l = 3n + 1$ .

$\beta\text{AsI}_3$  could then be described with such a structure. The iodine atoms are cubic closed packed and the AsE groups are located in the octahedral sites. As in  $\alpha\text{AsI}_3$ , two-thirds of the octahedra are filled within one layer formed by iodine atoms and with vacant contiguous layers.

Of course, the arsenic atoms are shifted toward one face of the octahedron as in  $\alpha\text{AsI}_3$  as to form molecular entities (Fig. 21). The ideal representation of such a structure is given in Fig. 22 in terms of  $\text{CrCl}_3$  structure type.

It should be noted that this  $\beta\text{AsI}_3$  structure exhibits a packing (CCP) less dense than the  $\alpha\text{AsI}_3$  form (HCP).

*Proposal for the mechanism of the phase transition.* The comparison of the two idealized Figs. 20 and 22 generates a simple proposal for the phase transition mechanism. The identity of the layers L1 (Figs. 19 and 21) should be mentioned. To get the  $\beta$  form from the  $\alpha$  form, layers L2 and L3 just have to be moved, and reversely. This scheme is indicated in Fig. 23 for the  $\alpha \rightarrow \beta$  transition:

(1) the layers L1 are kept in their original position

(2) the layers L2 are shifted by a vector with a  $2a/6$  modulus in the direction  $\{110\}$ , i.e.,  $2a/6 \{110\}$

(3) the layers L3 are shifted by a vector with a  $a/6$  modulus in the direction  $\{110\}$  and  $a\sqrt{3}/6$  in the direction  $\{110\}$ , i.e.:  $a/6 \{110\} + a\sqrt{3}/6 \{110\}$ .

After this operation, Fig. 22 is derived from Fig. 20. Such a mechanism requires only weak forces to act, as the layers L1,

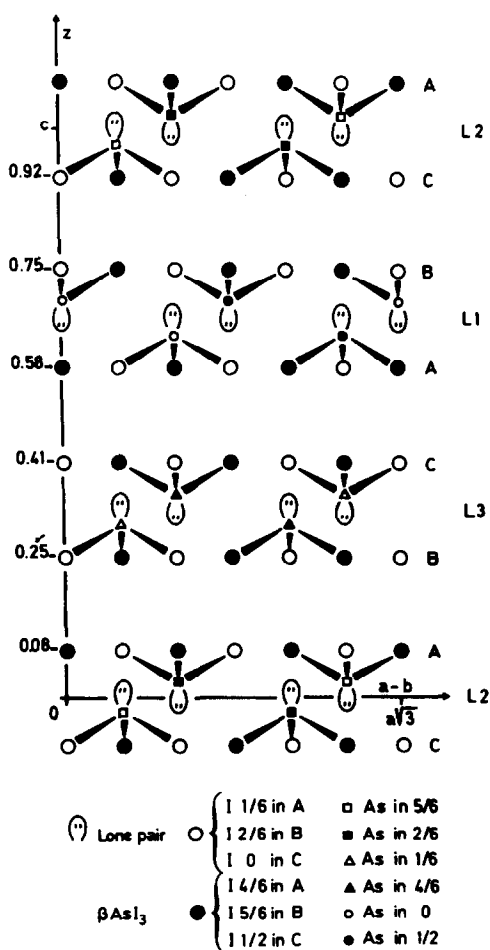


FIG. 21. The projection of the  $\beta\text{AsI}_3$  structure onto the plane (110) proposed after the  $\text{CrCl}_3$  structure type.

L2, and L3 are separated by vacant layers in which only van der Waals bonds exist. This driving force is readily obtained by gentle heating ca.  $110^\circ\text{C}$ , to transform  $\alpha \rightarrow \beta$ . After a few days at room temperature,  $\beta \rightarrow \alpha$ . The kinetics of such a transformation is accelerated greatly by heating the sample ca.  $80^\circ\text{C}$ .

*A possible high pressure form for  $M^*I_3$  ( $\alpha\text{AsI}_3$  structure type)?* In 1969, Andersson and Galy (54) proposed a mechanism, by simple cation rearrangement, to explain the way  $\text{TiO}_2(\text{I})$  (rutile) transforms into  $\text{TiO}_2(\text{II})$  ( $\alpha\text{PbO}_2$  structure type) under high pressure

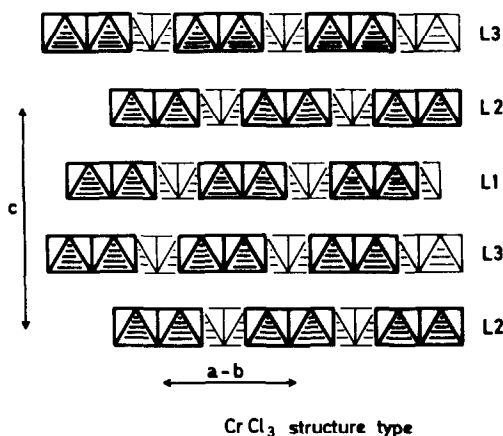


FIG. 22. Ideal representation of the  $\beta\text{AsI}_3$  structure with the  $\text{CrCl}_3$  structure type.

(Fig. 24). Such a mechanism was later confirmed in an experiment performed by Colaitis and Lecaille (55) who studied the transformation of  $\text{ReO}_2$  (rutile)  $\rightarrow$   $\text{ReO}_2$  ( $\alpha\text{PbO}_2$ ) using electron microscopy. Similar mechanisms were also successfully proposed to relate  $\text{Li}_2\text{ZrF}_6$ , trirutile, and  $\text{Na}_2\text{SiF}_6$  structure types, and to follow more easily the phase transitions of the compounds with the general formula  $\text{AB}_2\text{X}_6$  in a pressure-temperature phase diagram (56).

If pressure is applied on a triiodide with the  $\alpha\text{AsI}_3$  structure type, the repulsion between  $M^*-M^*$  may be increased because of a new repartition requiring the occupation of the empty layers in order to obtain the

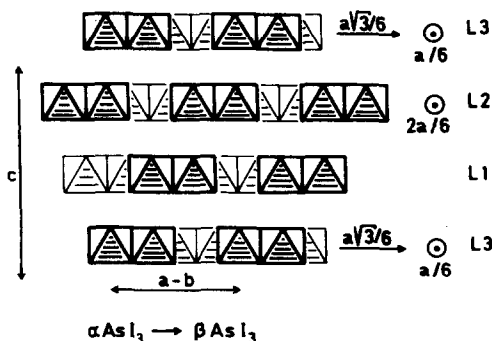


FIG. 23. Proposed mechanism for the  $\alpha \rightarrow \beta\text{AsI}_3$  transformation (projection onto the plane (110)).

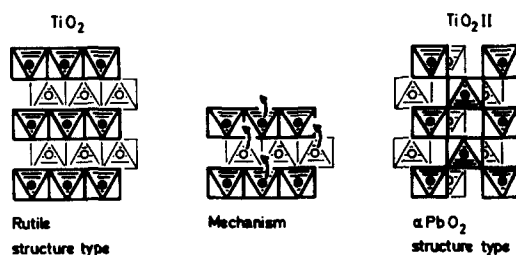


FIG. 24. The mechanism for the rutile  $\rightarrow$   $\alpha\text{PbO}_2$  transformation.

most important interatomic distances between  $M^*$  atoms. Such a possibility is illustrated by Fig. 25 where the arrows indicate the diffusion route of half the  $M^*$  atoms through the faces of iodine octahedra to occupy the neighboring empty layers. Such a mechanism would involve a cooperative phenomenon.

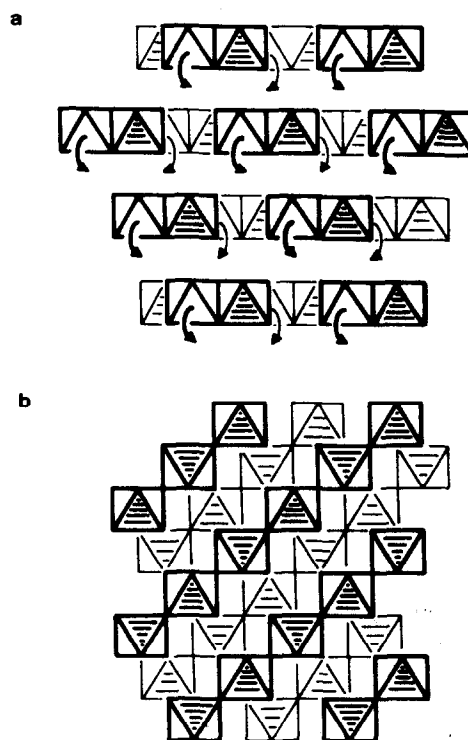


FIG. 25. Proposal for a possible high-pressure transformation of  $\alpha\text{AsI}_3$ : (a)  $\alpha\text{AsI}_3$ ; the diffusion way of the atoms is indicated by the arrows. (b) Hypothetical  $\text{AsI}_3$  high-pressure form isostructural with  $\text{VF}_3$ .

The resulting structure is actually simple, i.e., the  $VF_3$  type in which the filled octahedra share only corners.

#### IV. $M^*X_3$ Molecules and Lone Pair Stereochemistry

As already stated, the trihalides of the VA elements,  $M^*X_3$ , show molecules entities in the solid state. Their classification in structural types has been achieved, it is interesting to compare the typical bond lengths  $M^*-X$  and bond angles  $X-M^*-X$  of the molecules in their different states (gas, liquid, or solid).

Where available, the remarkable similarity of the data obtained by various techniques for a given  $M^*X_3$  molecules should be noted (Table IV).

In all these trihalides,  $M^*$  and its lone pair  $E$  are found in three main different coordination polyhedra ( $\alpha BiF_3$  is the only exception with BiE in a monocapped trigonal antiprism) (Fig. 26):

(1) C.N. 9, as a tricapped trigonal prism exemplified by the  $AsCl_3E$  case;

(2) C.N. 8, as a bicapped trigonal prism, for which a view of the  $PBr_3E$  entity is given;

(3) C.N. 6, as a trigonal antiprism, with the example of the octahedron of iodine in  $AsI_3E$ .

All the  $M^*-X$  distances included in these coordination polyhedra are summarized in Table V. Such a description, which accounts for all the anions surrounding  $M^*E$ , has been presented by Fourcade (57) during an original chemical investigation of anti-mony compounds.

As expected from VSEPR theory (5), the bond angles decrease when the electronegativity of  $M^*$  decreases ( $N \rightarrow Bi$ ) or when that of the ligands  $X$  increases ( $I \rightarrow F$ ).

#### Structural Evolution with the Nature of the Ligand $X$

The interaction between the lone pair borne by  $M^*$  and the nature of the anions

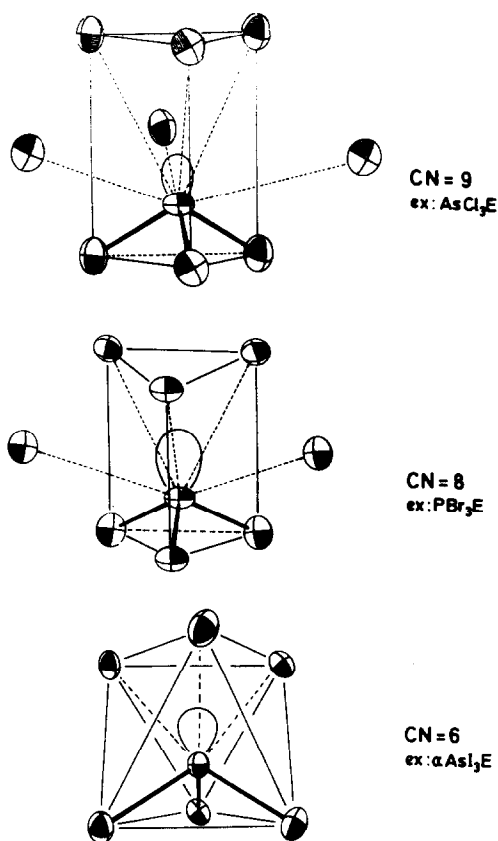


FIG. 26. Coordination polyhedra C.N. 9, 8, and 6.

which comprise the coordination polyhedra are illustrated by the arsenic trihalides (C.N. 9). Figure 27 clearly shows this structural evolution. When  $X$  represents the iodine atoms, the important size of the latter makes the "tricapped trigonal prism" as two octahedra sharing one face. When the size of  $X$  is smaller, i.e., with bromine and chlorine, the preceding polyhedron is transformed into a tricapped trigonal prism (compression along the pseudo threefold axis to make a regular trigonal prism with  $X_{A's}$  and  $X_{B's}$  anions, and with  $X_{C's}$  slightly displaced from the axis. Such a polyhedron surrounds the arsenic atom and its lone pair  $E$ . When fluorine is substituted, the trigonal prism is too small to allow  $AsE$  to align itself along its own threefold axis. The arse-



TABLE IV  
COMPARISON OF THE  $M^*X_3$  MOLECULAR DATA<sup>a</sup>

$M^*(VA)$	F			Cl			Br			I		
	$M-X$	$<X-M-X$	$X-X$	$M-X$	$<X-M-X$	$X-X$	$M-X$	$<X-M-X$	$X-X$	$M-X$	$<X-M-X$	$X-X$
N	ed	1.371	NF <sub>3</sub>	RX	1.75	106.8	2.82	NBr <sub>3</sub>				
P	ed	1.570	PF <sub>3</sub>	RX	2.034	100.1	3.119	PBr <sub>3</sub>	100.5	3.404	RX	2.463
				mw	2.040	100.3	3.132	ed	2.20	101	3.40	ed
As	RX mw	1.707 1.706	AsF <sub>3</sub>	RX	2.167	97.7	3.265	AsBr <sub>3</sub>	97.7	3.559	RX	2.591
				mw	2.161	98.7	3.279	ed	2.33	99.7	3.56	ed
Sb	RX	1.92	SbF <sub>3</sub>	RX	2.36	95.2	3.49	SbBr <sub>3</sub>	95	3.69	RX	2.87
				$\eta$ mw	2.35 2.325	96 99.5	3.49 3.549	$\alpha$ ed	2.50 2.51	95 97	3.69 3.76	ed
Bi	RX ed	2.50 2.48	BiF <sub>3</sub>									

<sup>a</sup> Bond lengths and interatomic distances (Å), bond angles (°). Methods: X-ray (RX), electron diffraction (ed), microwaves (mw), and neutrons (n).

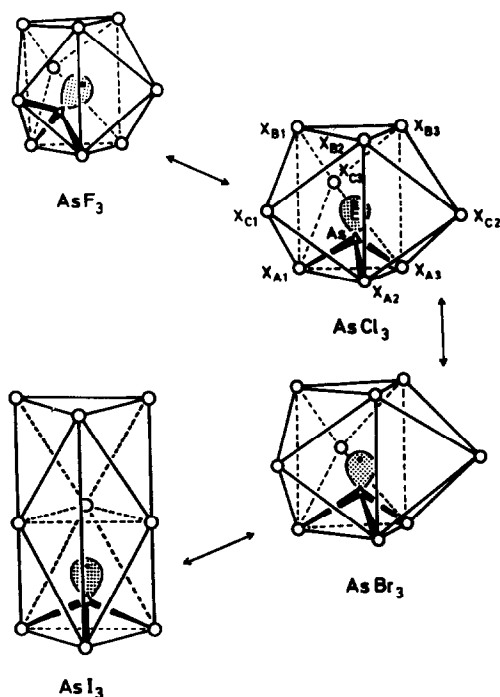


FIG. 27. Correlative evolution of both polyhedra and bonding distribution in the arsenic trihalides crystal structures.

nic atom is then pushed toward one of the capped square pyramids in order to leave as much space as possible in the polyhedron for the nonbonded electron pair and its sphere of influence.

#### Structural Behavior of the Elements $M^*$

The  $M^*$  cations behave like very big cations (i.e., plutonium in  $\text{PuBr}_3$ ) as are entities such as "PO" or "VO" (see  $\text{POX}_3$  ( $X = \text{Cl}, \text{Br}$ ) and  $\text{VOCl}_3$  structures), when involved in crystal structures.

This influence of the nature of  $M^*$  together with its lone pair is demonstrated by the triiodide structures. Indeed, the separation between iodine planes sandwiching  $M^*E$  entities ( $\text{PI}_3$ , 3.707 Å,  $\text{AsI}_3$ , 3.669 Å,  $\text{SbI}_3$ , 3.602 Å,  $\text{BiI}_3$ , 3.548 Å), clearly indicates that the sphere of influence more closely surrounds  $M^*$  when the size of the cation increases. The nature of  $M^*$  is also

important in the structural evolution of the trichlorides and tribromides. When  $Z$  increases, a deviation appears at the level of the arsenic compounds which crystallize with the  $\text{AsBr}_3$  structure type, whereas  $\text{NCl}_3$ ,  $\text{PX}_3$  and  $\text{BiX}_3$  crystallize with the  $\text{YF}_3$  structure type, and  $\text{SbX}_3$  is able to crystallize in both structure types. In the case of arsenic, the new molecular packing, with an unidimensional type, might be due to the  $d$  orbital filling, the direction  $M^*-E$  being more rigidly determined. Such a pattern is more vague for antimony which allows it to organize its molecular packing in both  $\text{YF}_3$  and  $\text{AsBr}_3$  ways. In the case of bismuth, the partial  $s$  character of the lone pair makes  $\text{BiX}_3$  crystallize again with the  $\text{YF}_3$  structure type. It is to be noted that the volume of  $\text{BiCl}_3$  is 10% smaller than that of the corresponding  $\text{SbCl}_3$ . Such an outstanding fact indicates that relativistic effects also are acting.

#### Evaluation of the Stereochemical Influence of the Lone Pair $E$

It is then tempting to apply the ideas and suggestions developed by Galy *et al.* (8) to determine the coordinates of the center  $E_s$  of the sphere of influence of the lone pair.  $E_s$  is situated at the apex of a trigonal pyramid based on the three halogens and including  $M^*$  and the centroid of the lone pair, designated by  $E_c$ , as shown in the scheme given in Fig. 28. Using such methods, the distances  $M^*-E_s$  were obtained. The resulting values were compared with the values obtained from a formula proposed in a previous paper by Galy and Enjalbert (20):

$$M^*-E_s = (a_e^2 - a_t^2/3)^{1/2} - (d_t^2 - a_t^2/3)^{1/2}$$

where  $a_t$  is the average value of the  $X-X$  distances in  $M^*X_3$ ,  $a_e = a_t/\sqrt{2} + 1.30$  Å (1.30 Å is the radius of the sphere of influence of  $E$  (7, 8)), and  $d_t$  is the mean distance  $M^*-X$ :  $N-E_s = 1.50$ ,  $P-E_s = 1.25$ ,  $As-E_s = 1.15$ ,  $Sb-E_s = 1.05$ , and  $Bi-E_s = 0.90$  Å.

TABLE V  
COVALENT AND SECONDARY BOND LENGTHS WITHIN THE COORDINATION POLYHEDRA CONTAINING  $M^*$

$M^*(VA)$	F	Cl	Br	I
	X			
N	NF <sub>3</sub> 1.73 (x2) 1.77 (x1) 3.30 (x2) 3.36 (x1) CN = 6	NCl <sub>3</sub> 1.79 (x2) 1.71 (x1) 3.92 (x2) 3.19 (x1) 3.96 (x2) CN = 8	NBr <sub>3</sub> 1.78 (x2) 1.70 (x1) 3.69 (x2) 3.63 (x1) 3.89 (x2) CN = 8	NI <sub>3</sub> 2.212 (x2) 2.216 (x1) 3.869 (x2) 3.754 (x1) 3.926 (x2) CN = 6
P	PF <sub>3</sub> 2.038 (x2) 2.026 (x1) 3.852 (x2) 3.612 (x1) 3.853 (x2) CN = 8	PCl <sub>3</sub> 2.038 (x2) 2.026 (x1) 3.852 (x2) 3.612 (x1) 3.853 (x2) CN = 8	PBr <sub>3</sub> 2.212 (x2) 2.216 (x1) 3.869 (x2) 3.754 (x1) 3.926 (x2) CN = 6	PI <sub>3</sub> 2.212 (x2) 2.216 (x1) 3.869 (x2) 3.754 (x1) 3.926 (x2) CN = 6
As	AsF <sub>3</sub> 1.699 (x1) 1.700 (x1) 1.721 (x1) 2.886 (x1) CN = 9	AsCl <sub>3</sub> 2.940 (x1) 3.052 (x1) 3.627 (x1) 3.184 (x1) 3.599 (x1) 2.886 (x1) CN = 9	AsBr <sub>3</sub> 3.773 (x1) 3.692 (x1) 3.727 (x1) 2.354 (x1) 2.354 (x1) 2.384 (x1) CN = 9	AsI <sub>3</sub> 3.738 (x1) 3.863 (x1) 3.869 (x1) 3.947 (x1) 4.170 (x1) 4.210 (x1) CN = 6
Sb	SbF <sub>3</sub> 1.94 (x2) 1.90 (x1) 2.60 (x2) 2.63 (x1) 3.93 (x2) CN = 8	SbCl <sub>3</sub> 2.368 (x2) 2.340 (x1) 3.457 (x2) 3.609 (x1) 3.736 (x2) CN = 8	SbBr <sub>3</sub> 3.79 (x1) 3.81 (x1) 3.75 (x1) 3.86 (x1) 4.09 (x1) 4.18 (x1) CN = 9	SbI <sub>3</sub> 2.51 (x2) 2.47 (x1) 3.70 (x2) 3.82 (x1) 3.88 (x2) CN = 8
Bi	BiF <sub>3</sub> 2.49 (x3) 2.39 (x1) 2.59 (x3) 2.69 (x1) CN = 8	BiCl <sub>3</sub> 2.52 (x2) 2.47 (x1) 3.22 (x2) 3.26 (x1) 3.43 (x2) CN = 8	BiBr <sub>3</sub> 2.52 (x2) 2.47 (x1) 3.22 (x2) 3.26 (x1) 3.43 (x2) CN = 8	BiI <sub>3</sub> 3.09 (x6) CN = 6

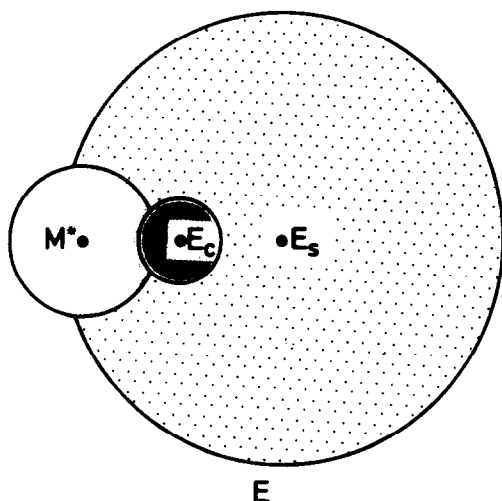


FIG. 28. Schematic representation of the atom  $M^*$ , with the centroid  $E_c$  and the sphere of influence  $E_s$  of the lone pair  $E$ .

A reasonable agreement is found for these calculated  $M^*-E_s$  distances with those already published on arsenic, antimony, and bismuth (8). Such values derived by applying geometrical principles and by averaging some distances in real crystal structures are probably correct within 0.1 Å.

Another way to obtain these  $M^*-E_s$  distances was by calculating the coordinates of the bary-center of the coordination polyhedron around  $M^*$  where  $E_s$  is supposed to be located.

Good agreement was found with the above values, the results being generally smaller by less than 0.1 Å, but this calculation does not work for the iodides, since their large size and high polarizability make them deviate.

The observed  $M^*-E_s$  values are listed in Table VI together with those corresponding to the  $M^*$  elements.

From these remarks, it is concluded that: (1) secondary interactions between  $M^*$  and  $X$  anions of other molecules exhibit shorter

TABLE VI  
 $M^*-E_s$  DISTANCE (Å)

		N(III)			
		1.50			
		P(III)	S(IV)	Cl(V)	
		1.25	1.46	1.50	
Ga(I)	Ge(II)	As(III)	Se(IV)	Br(V)	
0.95	1.05	1.26	1.22	1.47	
In(I)	Sn(II)	Sb(III)	Te(IV)	I(V)	Xe(VI)
0.86	0.95	1.06	1.25	1.23	1.49
Tl(I)	Pb(II)	Bi(III)	Po(IV)		
0.69	0.86	0.98	1.06		

distances when  $Z$  increases from N to Bi atoms in one column of the periodic table; correlatively, the  $M^*-E_s$  values decrease; (2) in other respects, when extending the fundamental results obtained by Schmiedekamp *et al.* (6) it appears that the centroid  $E_c$  of the lone pair moves away from the nucleus when passing from nitrogen to bismuth (example:  $N-E_c = 0.38$  Å for  $NF_3$ ,  $P-E_c = 0.57$  Å for  $PF_3$  (6)).

A possible explanation might lie in the ability of the lone pair, when its centroid is close enough to the nucleus, to screen it and then to make difficult the acting of the secondary interactions in this direction opposite the covalent bond.

When going from top to bottom of the VA column, the nucleus of  $M^*$  increases its size and the centroid of the lone pair is somewhat moved away from its center. The secondary interactions  $M^*-X$  can then be established attracting the neighbor molecules to  $M^*$  and, by the way, virtually bringing closer to  $M^*$  the sphere of influence, which consequently makes  $M^*-E_s$  decrease.

With the same basic hypothesis, this explanation can be extended to one row of the periodic table. For example, increasing values of  $M^*-E_s$  have also been found in the row In(I) to Xe(VI) (Table VI; Ref. (8)). The charge of the nucleus drastically in-

creases from indium to xenon and the nucleus should then attract more and more the electron pair, with correlatively expanding  $M^*-E_s$  distances.

Thus the "diagonal rule" of the periodic table is followed and it is readily seen that the  $Xe-E_s$  distance is very close to that of  $N-E_s$ . Looking at the  $XeO_3$  structure, in which the trigonal prism packing is very distorted compared to the "isostructural" one of  $AsF_3$  (see Fig. 2), it is tempting to consider that if we had a crystal of  $PF_3$  (possible) its crystal structure should be a bit more distorted than that of  $AsF_3$  and that of  $NF_3$  very close to  $XeO_3$  (. . . but . . .  $NF_3$  is solid at a temperature below 66 K).

## Appendix I

### Crystal Structure of $AsF_3$

$AsF_3$  is a liquid at room temperature. It is very sensitive to moisture and very reactive. A droplet was inserted and sealed in a Teflon capillary. The crystal was grown *in situ* on the CAD4 automatic diffractometer using the techniques detailed in Enjalbert's thesis (10). The data were collected using  $Mo K\alpha$  radiation.

Some crystallographic data are reported in Tables I, IV, and V of the paper and the atom positions and thermal parameters in Table IA.

TABLE IA

$AsF_3$	(193 K)	$a = 7.018(5)$	$b = 7.315(6)$	$c = 5.205(6)$ (Å)		
	$Pn2_1a$	$x$	$y$	$z$	$B_{eq}^b$ (Å <sup>2</sup> )	
As		0.2862(2)	$\frac{1}{4}^a$	0.0338(2)	1.7(1)	
F1		0.0736(17)	0.3065(19)	-1.005(20)	4.6(2)	
F2		0.3077(18)	0.0490(17)	-0.1260(26)	3.7(2)	
F3		0.1792(16)	0.1493(16)	0.2956(20)	3.3(2)	
	$\beta_{11}(\times 10^4)$	$\beta_{22}$	$\beta_{33}$	$\beta_{12}$	$\beta_{13}$	$\beta_{23}$
As	75(3)	103(3)	145(5)	-35(4)	0(2)	28(7)
F1	193(23)	299(41)	330(37)	20(26)	-115(30)	48(3)
F2	241(32)	164(22)	262(41)	-11(21)	61(29)	-79(27)
F3	217(26)	162(20)	189(37)	2(20)	84(24)	8(21)

<sup>a</sup> Polar space group,  $y$  is arbitrarily fixed.

<sup>b</sup>  $B_{eq} = \frac{1}{3} \sum (\mathbf{a}_i \cdot \mathbf{a}_j) \beta_{ij}$ ; factor temperature form:  $\exp(-(\beta_{11}h^2 + \beta_{22}k^2 + \beta_{33}l^2 + 2\beta_{12}hk + 2\beta_{13}hl + 2\beta_{23}kl))$ .

## Appendix II

### Crystal Structure of $AsCl_3$

A small drop of liquid  $AsCl_3$  was placed in a Lindemann capillary and a crystal was

obtained using the methods already described (10).

Complete data are given in Tables I, IV, and V of this paper and the detailed atomic parameters are listed in Table IIA.

TABLE IIA

AsCl <sub>3</sub>	(253 K)	$a = 9.466(3)$		$b = 11.335(3)$		$c = 4.289(2)$ (Å)	
	$P2_12_12_1$	$x$		$y$		$z$	$B_{eq}^2$ (Å <sup>2</sup> )
As		0.3010(1)		0.2879(1)		0.4891(2)	2.62(6)
Cl1		0.2966(2)		0.1236(1)		0.2321(6)	3.2(1)
Cl2		0.1347(2)		0.3792(2)		0.2380(7)	3.4(1)
Cl3		0.4782(2)		0.3676(2)		0.2490(8)	3.4(1)
	$\beta_{11}(\times 10^4)$	$\beta_{22}$	$\beta_{33}$	$\beta_{12}$	$\beta_{13}$	$\beta_{23}$	
As	84(1)	61(1)	231(3)	3(1)	3(3)	-2(2)	
Cl1	90(2)	42(1)	578(12)	4(2)	-7(7)	6(4)	
Cl2	76(2)	62(2)	564(14)	18(2)	11(6)	3(6)	
Cl3	77(2)	62(2)	564(13)	-11(2)	-14(6)	11(6)	

<sup>a</sup>  $B_{eq} = \frac{1}{3} \sum (\mathbf{a}_i \cdot \mathbf{a}_j) \beta_{ij}$ ; factor temperature form:  $\exp(-(\beta_{11}h^2 + \beta_{22}k^2 + \beta_{33}l^2 + 2\beta_{12}hk + 2\beta_{13}hl + 2\beta_{23}kl))$ .

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